



This open access document is posted as a preprint in the Beilstein Archives at <https://doi.org/10.3762/bxiv.2024.72.v1> and is considered to be an early communication for feedback before peer review. Before citing this document, please check if a final, peer-reviewed version has been published.

This document is not formatted, has not undergone copyediting or typesetting, and may contain errors, unsubstantiated scientific claims or preliminary data.

**Preprint Title** Nanoscale capacitance spectroscopy based on multi-frequency electrostatic force microscopy

**Authors** Pascal N. Rohrbeck, Lukas D. Cavar, Franjo Weber, Peter G. Reichel, Mara Niebling and Stefan A. L. Weber

**Publication Date** 20 Dez. 2024

**Article Type** Full Research Paper

**Supporting Information File 1** H-SCM Publication SI.zip; 64.2 MB

**ORCID® IDs** Pascal N. Rohrbeck - <https://orcid.org/0000-0002-1514-6008>



License and Terms: This document is copyright 2024 the Author(s); licensee Beilstein-Institut.

This is an open access work under the terms of the Creative Commons Attribution License (<https://creativecommons.org/licenses/by/4.0>). Please note that the reuse, redistribution and reproduction in particular requires that the author(s) and source are credited and that individual graphics may be subject to special legal provisions.

The license is subject to the Beilstein Archives terms and conditions: <https://www.beilstein-archives.org/xiv/terms>.

The definitive version of this work can be found at <https://doi.org/10.3762/bxiv.2024.72.v1>

# 1 **Nanoscale capacitance spectroscopy based on multi-frequency electro-** 2 **static force microscopy**

3 Pascal N. Rohrbeck<sup>1,2</sup>, Lukas D. Cavar<sup>1,3</sup>, Franjo Weber<sup>1,2</sup>, Peter G. Reichel<sup>1</sup>, Mara Niebling<sup>1,3</sup>  
4 and Stefan A. L. Weber<sup>\*1,3,4</sup>

5 Address: <sup>1</sup>Max Planck Institute for Polymer Research, Ackermannweg 10, 55128 Mainz, Ger-  
6 many; <sup>2</sup>Department of Chemistry, University of Mainz, Duesbergweg 10-14, 55128 Mainz, Ger-  
7 many; <sup>3</sup>Department of Physics, University of Mainz, Staudingerweg 7, 55128 Mainz, Germany and  
8 <sup>4</sup>Institute for Photovoltaics, University of Stuttgart, Pfaffenwaldring 47, 70569 Stuttgart, Germany

9 Email: Stefan A. L. Weber - stefan.weber@mpip-mainz.mpg.de; Stefan.Weber@ipv.uni-  
10 stuttgart.de

11 \* Corresponding author

## 12 **Abstract**

13 We present Heterodyne Scanning Capacitance Microscopy (H-SCM) as a novel multi-frequency  
14 electrostatic force microscopy method for nanoscale capacitance characterization. Next to a high  
15 spatial resolution, the key advantage of the multi-frequency approach of H-SCM is that it allows for  
16 operation at almost arbitrary frequencies, enabling the measurement of the local dielectric function  
17 over a wide range of frequencies. We demonstrate the reliable operation of H-SCM using standard  
18 Atomic Force Microscopy (AFM) equipment plus an external lock-in amplifier up to a frequency of  
19 5 MHz. Our results show a significant reduction of signal background, resulting in higher locality  
20 of the measurements with less cross-talk. Combined with improved models for the tip-sample ca-  
21 pacity, H-SCM will pave the way for quantitative studies of dielectric effects in nanoscale systems  
22 in materials science, biology, and nanotechnology.

## 23 **Keywords**

24 Atomic force microscopy; capacity gradients; dielectric constant; dielectric spectroscopy; hetero-

25 dyne frequency mixing; Kelvin Probe Force Microscopy; quantitative force spectroscopy; scanning  
26 capacitance microscopy; multi frequency AFM;

## 27 **Introduction**

28 Technological progress in fields including electronics, energy storage, photonics, and biomed-  
29 cal devices would not have been possible without the development of new materials. Progress in  
30 these areas requires a detailed understanding of material properties, particularly at the nanoscale,  
31 where phenomena such as quantum confinement, interface effects, and defect dynamics play a  
32 critical role. Innovations in characterization techniques have enabled researchers to explore these  
33 properties with unprecedented precision, paving the way for the design of materials with tailored  
34 functionalities[1-6].

35 Dielectric properties are fundamental for understanding the behavior and performance of various  
36 material systems, as they directly influence charge storage, polarization, and energy dissipation  
37 mechanisms. For instance, in microelectronic devices, high- $\kappa$  dielectric materials such as  $\text{HfO}_2$  and  
38  $\text{ZrO}_2$  are critical for minimizing leakage currents and enhancing gate capacitance in transistors[7-  
39 9]. In energy storage systems, the dielectric constants of polymer-ceramic composites determine  
40 the efficiency and reliability of capacitors[10]. Similarly, in next-generation photovoltaic devices,  
41 the dielectric properties of absorber layers, such as lead-halide perovskites, affect carrier recombi-  
42 nation and electric field distribution, thereby influencing power conversion efficiency[11].

43 At the nanoscale, the importance of dielectric properties becomes even more pronounced. Many  
44 advanced materials exhibit nanoscale structural heterogeneity, where quantum confinement, phase  
45 composition, and interfacial effects cause significant deviations in dielectric behavior compared to  
46 bulk materials[12,13]. These nanoscale variations influence key properties such as charge trans-  
47 port, polarization dynamics, and defect distributions, directly impacting the performance of micro-  
48 electronic and energy systems[14,15]. Understanding these effects requires correlating nanoscale  
49 dielectric properties with structural and morphological features.

50 Scanning probe techniques have revolutionized nanoscale material characterization. Since the in-

51 vention of Scanning Tunneling Microscopy (STM)[16] and Atomic Force Microscopy (AFM)[17],  
52 various electric force-based methods have emerged to study materials like perovskite solar cells[18-  
53 20] and Li-ion batteries[21-23]. AFM enables simultaneous acquisition of topographic and elec-  
54 tronic data by applying AC or DC voltages across the tip-sample gap, allowing the detection of  
55 capacitive forces[24,25] or contact potential difference (CPD)[18]. Its exceptional spatial reso-  
56 lution, ranging from sub-micron[24,26] to atomic scales[27,28], makes AFM a powerful tool for  
57 nanoscale analysis.

58 Scanning capacitance microscopy (SCM) is another widely used technique for capacitance mea-  
59 surements. SCM quantifies intrinsic material properties, such as film thickness[29,30] and dielec-  
60 tric constants[30,31], with superior spatial resolution compared to conventional methods like el-  
61 lipsometry or reflectance spectroscopy[30]. However, existing techniques face limitations due to  
62 nonlocal stray capacitances[32] and reliance on external cables and sensors, which compromise  
63 measurement accuracy and resolution[33,34].

64 To address these challenges, we present a novel, multi-frequency AFM-based method for nanoscale  
65 capacitance characterization. Our approach measures the second capacitance gradient ( $\propto \frac{\partial^2 C}{\partial z^2}$ ),  
66 enhancing localization by minimizing stray capacitance contributions[35]. This method enables  
67 high-frequency capacitance gradient spectroscopy without requiring specialized equipment beyond  
68 a lock-in amplifier.

69 The following sections introduce the theoretical framework of multi-frequency Electrostatic Force  
70 Microscopy (EFM), demonstrate its resolution enhancement experimentally, and validate its spec-  
71 troscopic capabilities by measuring nanoscale dielectric properties of microfabricated SiO<sub>2</sub> sam-  
72 ples. Finally, we compare its performance with established techniques through capacitance imaging  
73 of a Perfluoroalkyl-Alkane F(CF<sub>2</sub>)<sub>14</sub>(CH<sub>2</sub>)<sub>20</sub>H (F14H20) sample.

## 74 Theory

### 75 Multi-frequency Electrostatic Force Microscopy

76 The electrostatic force  $F_{\text{ES}}$  between tip and sample can be understood in terms of the gradient of  
77 the energy,  $W_C$ , stored in the tip-sample capacitor  $C$  with respect to the tip-sample separation  $z$ , as  
78 given by

$$79 \quad F_{\text{ES}} = \frac{\partial W_C}{\partial z} = \frac{1}{2} \cdot \frac{\partial C}{\partial z} \cdot V_{\text{tip-sample}}^2, \quad (1)$$

80 where  $V_{\text{tip-sample}}$  specifies the electrical voltage across the tip-sample gap. In conventional EFM  
81 with single-frequency excitation,  $V_{\text{tip-sample}}$  is given by eq. (2).

$$82 \quad V_{\text{ES}} = V_{\text{DC}} - V_{\text{CPD}} + V_{\text{AC}} \cdot \sin(\omega_e \cdot t)[18]. \quad (2)$$

83 with  $V_{\text{DC}}$  the direct current (DC)-Voltage offset applied to the tip,  $V_{\text{AC}}$  the alternating current (AC)  
84 voltage amplitude with the frequency  $\omega_{\text{AC}}$  at a certain time  $t$  and  $V_{\text{CPD}}$  the contact potential differ-  
85 ence (CPD), which corresponds to the difference in tip and sample work function[18]. Inserting  
86 equation eq. (2) into equation eq. (1), we obtain the following expression:

$$87 \quad F_{\text{ES}} = \frac{1}{2} \frac{\partial C}{\partial z} \left( (V_{\text{DC}} - V_{\text{CPD}})^2 + \frac{V_{\text{AC}}^2}{2} \right) \quad (3a)$$

$$88 \quad + \frac{\partial C}{\partial z} (V_{\text{DC}} - V_{\text{CPD}}) V_{\text{AC}} \sin(\omega_e t) \quad (3b)$$

$$89 \quad + \frac{\partial C}{\partial z} \frac{V_{\text{AC}}^2}{4} \cos(2 \omega_e t) \quad (3c)$$

90 Alongside a static component in eq. (3a), the electrostatic force has periodic time-dependent  
91 components at frequencies  $\omega_e$  and  $2\omega_e$  which corresponds to eqs. (3b) and (3c), respectively. In  
92 the case of an oscillating AFM tip, the tip-sample distance  $z$  and thereby the tip-sample capaci-  
93 tance and its gradients is changing periodically. This periodic fluctuation of the capacity gradient  
94  $C'(t) = \frac{\partial C}{\partial z}(t)$  adds an additional dynamic component to eq. (3). Using a Fourier expansion for the

95 capacitance gradient  $C'(t) = \frac{\partial C}{\partial z}(t)$  yields[18]:

$$96 \quad C'(t) = C'(z_0) + C''(z_0) \cdot A_m \cdot \cos(\omega_m t) + \dots \quad (4)$$

97 By inserting eq. (4) into eq. (3), we find that frequency mixing between  $C'(t)$  and the electrostatic  
98 excitation leads to sidebands at frequencies  $\omega_{SB,1} = (\omega_m \pm \omega_{AC})$  and  $\omega_{SB,2} = (\omega_m \pm 2\omega_{AC})$  next  
99 to the the mechanical oscillation at  $\omega_m$ [18]. The amplitude of the first harmonic frequency com-  
100 ponents is used in conventional amplitude modulation (AM) and sideband or heterodyne Kelvin  
101 Probe Force Microscopy (KPFM)[18]. The second harmonic signals are proportional to the local  
102 capacity gradients, providing information about the local tip-sample capacitance. To ensure a suf-  
103 ficient signal-to-noise ratio, the resulting frequencies should coincide with one of the cantilever's  
104 resonance frequencies, limiting the choice of excitation frequencies.

105 We can avoid this limitation by using a multi-frequency excitation approach. With a double-  
106 frequency excitation, we can write the tip-sample voltage as

$$107 \quad V_{\text{tip-sample}} = V_{AC,1} \cdot \sin(\omega_{e,1}) + V_{AC,2} \cdot \sin(\omega_{e,2}) \quad (5)$$

108 In the case of two drives with identical amplitude  $V_{AC,1} = V_{AC,2} = \frac{V_{AC}}{2}$ , eq. (5) can be rearranged as

$$109 \quad V_{\text{tip-sample}} = V_{AC} \cdot \sin\left(\frac{\omega_{e,1} - \omega_{e,2}}{2}t\right) \cdot \sin\left(\frac{\omega_{e,1} + \omega_{e,2}}{2}t\right). \quad (6)$$

110 Thus, the waveform can be viewed as a high-frequency oscillation at  $(\omega_{e,1} + \omega_{e,2})/2$  with a low-  
111 frequency amplitude modulation at frequency  $(\omega_{e,1} - \omega_{e,2})/2 = \Delta\omega_e/2$ . This effect is also known  
112 as "beating" and is utilized in the AFM context for example in intermodulation AFM[36-39].

113 By inserting eq. (4) and eq. (6) in eq. (3), we obtain the full expression for the electrostatic force.

114 Here, we will focus on the DC force component in eq. (3a) and set  $V_{\text{DC}} - V_{\text{CPD}} = \Delta$ :

$$115 \quad F_{\text{DC}} = \frac{1}{2} (C' + C'' A_m \sin(\omega_m t) + \dots) \cdot \left[ \Delta^2 + \frac{V_{\text{AC}}^2}{2} \sin^2(\Delta\omega_e/2t) \right]$$

$$116 \quad = \frac{1}{2} C' \left[ \Delta^2 + \frac{V_{\text{AC}}^2}{4} \right] + \frac{1}{8} C' V_{\text{AC}}^2 \cos(\Delta\omega_e t) \quad (7a)$$

$$117 \quad + \frac{1}{2} C'' A_m \left[ \Delta^2 + \frac{V_{\text{AC}}^2}{4} \right] \sin(\omega_m t) + \frac{1}{16} C'' A_m V_{\text{AC}}^2 \sin((\omega_m \pm \Delta\omega_e t)) \quad (7b)$$

118 In addition to a static force term identical to eq. (3a), eq. (7a) contains a term proportional to  $C'$  at  
 119 frequency  $2\omega_{\text{mod}} = \Delta\omega$ . This force has been used for AM based dielectric spectroscopy[31,40-46].  
 120 The second term eq. (7b) contains a force component at the mechanical drive frequency  $\omega_m$  and at  
 121 a sideband frequency  $\omega_m \pm 2\omega_{\text{mod}}$ . The latter one is independent of the local CPD, making it  
 122 interesting for dielectric measurements. As the magnitude of this force component depends on  $C''$ ,  
 123 we can expect a superior lateral resolution through a reduction of long-ranged force contributions  
 124 from tip cone and cantilever. As in the case of conventional EFM, signal-to-noise is greatly im-  
 125 proved by choosing  $\Delta\omega_e$  such that one of the induced sidebands falls on one of the cantilever's me-  
 126 chanical resonances. We call this method heterodyne Scanning Capacitance Microscopy (H-SCM).  
 127 To calculate the second capacitance gradient, we need to calculate the electrostatic force from the  
 128 detected amplitude signal,  $A_{\text{det}}$ , taking into account the cantilever's frequency-dependent spring  
 129 constant or transfer function,  $k(\omega)$ .

$$130 \quad \frac{\partial^2 C}{\partial z^2} = C'' = \frac{16 A_{\text{det}} \cdot k(\omega)}{A_m \cdot V_{\text{AC}}^2} \quad (8)$$

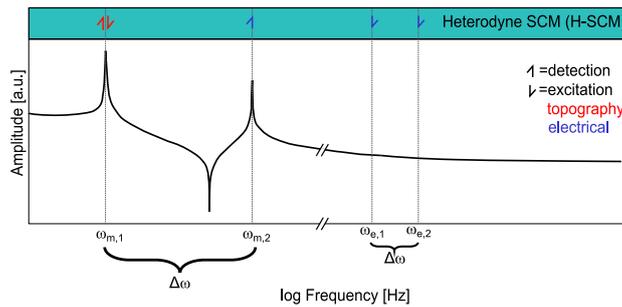
131 Interestingly, the forces in eq. (7b) are only dependent on the frequency difference,  $\Delta\omega_e$ , of the  
 132 electrical drive frequencies. Thus, the experiments can be performed at almost arbitrarily high AC  
 133 frequencies. The lower limit for the frequency range is given by the second resonance of the can-  
 134 tilever. Towards higher frequencies, the impedance of the electrical connection will introduce a  
 135 damping of the excitation signal that has to be considered in eq. (8). By using appropriate means of  
 136 coupling the electrical excitation into the tip-sample gap, experiments at microwave or even at opti-

137 cal frequencies are possible. In our setup, the two excitation frequencies can be varied in frequency  
 138 from  $\approx 600$  kHz up to at least 50 MHz, limited by the bandwidth of the lock-in amplifier. To reach  
 139 a nanoscale sensitive measurement of the dielectric constant in media besides air, a detection at  
 140 higher excitation frequencies in the MHz regime is strictly necessary[47].  
 141 The indirect detection of local capacity variations by means of an electrstatic force has the advan-  
 142 tage that it does not require additional devices for the measurement except for the lock-in ampli-  
 143 fier (LIA) similar to the work of Gramse et al.[48]. Nevertheless, to quantify the total tip-sample  
 144 capacitance will require varying the distance, e.g. by force-distance spectroscopy.

## 145 Methods

### 146 Heterodyne Scanning Capacitance Microscopy (H-SCM) to measure the sec- 147 ond capacitive gradient $C''$

148 We perform heterodyne Scanning Capacitance Microscopy (H-SCM) using a conductive AFM can-  
 149 tilever in tapping mode with a mechanical drive near the fundamental cantilever eigenmode  $\omega_{m,1}$   
 150 with a mechanical amplitude  $A_m$ . Additionally, we apply two high-frequency electrical excitations  
 151 of identical magnitude ( $V_{AC,1} = V_{AC,2}$ ) at the frequencies  $\omega_{e,1}$  and  $\omega_{e,2}$  (see eq. (5)). A schematic of  
 152 the excitation frequencies is shown in Figure 1.



**Figure 1:** Schematic illustration of the excitation and detection frequencies in H-SCM. The lower part shows the transfer function of the cantilever, where the amplitude is plotted vs the logarithmic angular frequency. The upper part shows the excitation frequencies ( $\downarrow$ ) and the detection frequencies ( $\uparrow$ ) of the applied frequencies. The red arrow corresponds to topography- and the blue arrow to the electrical signal. Representation of Fig. 1 was inspired by [18,26]. A comparison of Heterodyne Kelvin Probe Force Microscopy (H-KPFM) and H-SCM can be found in Supporting Information File 1.

153 The electrical detection frequency (300 - 420 kHz) is several hundred kilohertz away from the fre-  
 154 quency of mechanical oscillation (65 - 80 kHz), effectively reducing crosstalk between the topo-  
 155 graphical and capacitive images.

156 We select the excitation frequencies to lie at the  $n$ th and the  $n + 1$ th multiple of the frequency gap  
 157  $\Delta\omega = (\omega_{m,2} - \omega_{m,1})$  (see Figure 1), respectively. We then use lock-in detection to measure the  
 158 induced mechanical excitation at exactly at the second harmonic of the cantilever ( $\omega_{m,2}$ ).

### 159 **Single-frequency Electrostatic Force Microscopy to measure the first capaci-** 160 **tive gradient $C'$**

161 To obtain a quantitative comparison of the signal contributions in the signals based on the first and  
 162 the second capacitance gradient we performed single-frequency excitation EFM (SF-EFM) mea-  
 163 surements as comparison to the multi-frequency approach described above. In the fixed-frequency  
 164 configuration, we use lock-in amplification to detect the second harmonic force component at  $2\omega_e$   
 165 induced by a single-frequency ( $\omega_E$ ) stimulus (see eq. (3c)).

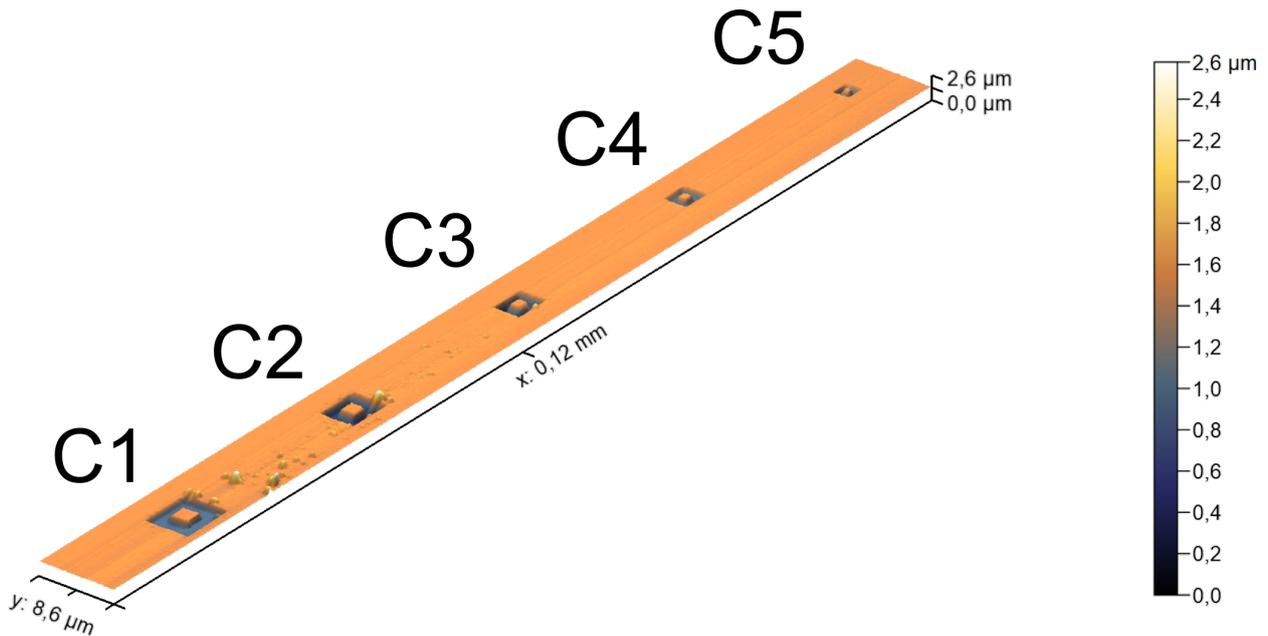
166 To enhance the signal we select  $\omega_E$  such that  $2\omega_E$  coincides with the second resonance of the can-  
 167 tilever ( $2\omega_E = \omega_{m,1} + \omega_E = \omega_{m,2}$ ). We relate the numerical value of the capacitance gradient to the  
 168 detected amplitude via the cantilever's frequency-dependent spring constant  $k(\omega)$  by:

$$169 \quad \frac{\partial C}{\partial z} = C' = \frac{4 A_{\text{det}} \cdot k(\omega)}{V_{\text{AC}}^2} \quad (9)$$

170 For the variable-frequency detection of  $C'$  we apply two AC voltages of the same magnitude  
 171 ( $V_{\text{AC},1} = V_{\text{AC},2}$ ) at frequencies  $n$  and  $(n + 1)$  times the second resonance frequency  $\omega_{m,2}$ . According  
 172 to eq. (7a), this will excite an oscillation at  $\omega_{m,2}$  with an amplitude proportional to  $C'$ .

### 173 **Silicon micro capacitors**

174 To compare and verify the  $C'$  and  $C''$  signal dependency as a function of  $z$  during several force-  
 175 distance curves from the literature with our data we performed experiments on one of the prepared  
 176 "microcapacitors" you can see in Figure 2.



**Figure 2:** Topography of the five different capacitors C1 to C5 that were produced to have specific capacitors with known capacity. The picture was conducted with the  $\mu$ masch's HQ:NSC18/Pt Cantilever and analysed with Gwyddion 2.61.

177 The microcapacitors were produced from scratch by focused ion beam (FIB) milling on a Silicon  
 178 wafer which was thermally modified so that it has a 300 nm layer of  $\text{SiO}_2$  on it. After that the Sil-  
 179 icon wafer was placed in the Pt sputter machine and sputtered 14 nm on it. The final substrate had  
 180 the following layers of Si/300 nm $\text{SiO}_2$ /14 nmPt. A mask was used to mill a trench into the sample  
 181 until the underlaying Si substrate was visible so that the microcapacitors stick out of surface. This  
 182 gives a defined nano-structure with known capacitance.

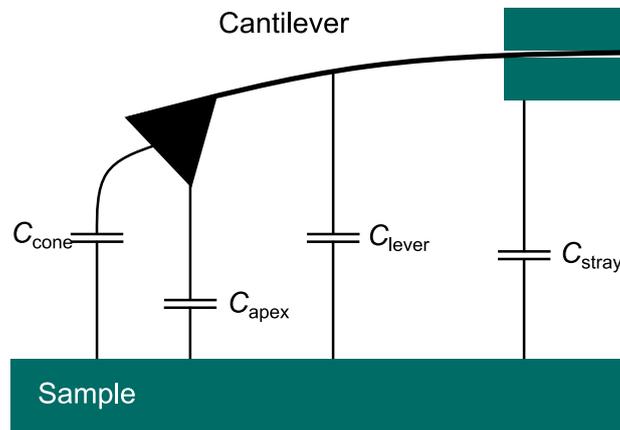
## 183 Results and Discussion

184 To investigate whether the  $C''$ -sensitive detection leads to an improved spatial resolution of  
 185 H-SCM as compared to conventional methods, we calculate the distance-dependence of the  
 186 first- and second-order capacity gradients in an ideal cantilever. We compare our calculations  
 187 to experimentally-obtained force-distance curves. We then show the first practical examples of  
 188 high-frequency capacitive spectra obtained by this method in etched  $\text{SiO}_2$  microcapacitors, along

189 with high-resolution high-frequency capacitance-images obtained over self-assembled molecular  
190 Perfluoroalkyl-Alkane  $F(CF_2)_{14}(CH_2)_{20}H$  (F14H20).

### 191 **Tip-sample capacitance**

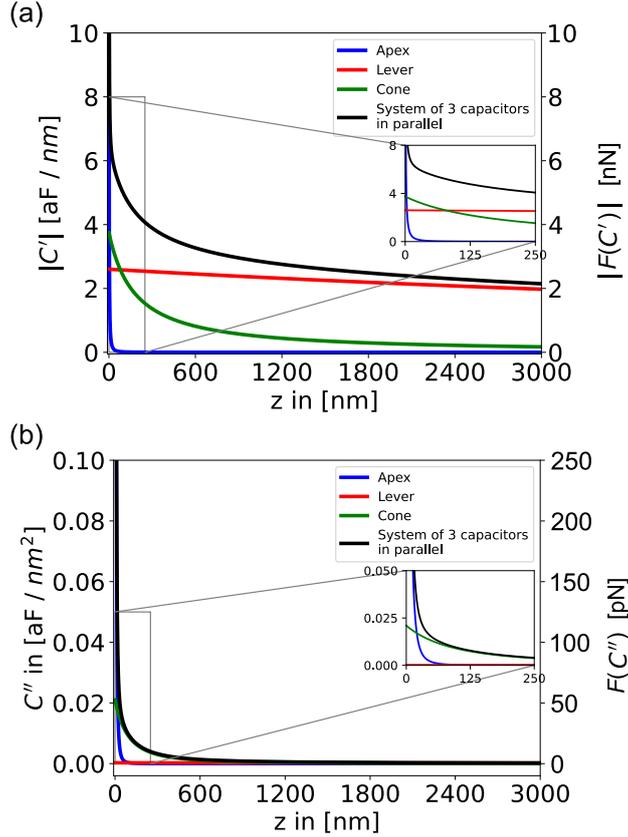
192 The total capacitance between the sample and the cantilever consists of contributions from the tip  
193 apex, tip cone, lever and some additional stray capacitance caused by the signal cables in the AFM  
194 head (Figure 3). Whereas the apex capacitance contains the desired local information, the stray  
195 capacitance from cone, lever and cables produces a background signal that effectively reduces the  
196 lateral resolution of the local capacitance measurement. Practically, these signal contributions can  
197 be discerned by their respective distance dependence.



**Figure 3:** Schematic illustration of tip apex, tip cone, lever and stray capacitance. The contribution of the tip-apex contains the most localized part of the overall capacitance signal. The mesoscopic tip cone and the macroscopic cantilever, on the other hand, contribute to long-ranged stray capacitance, effectively delocalizing the signal.

198 To further investigate this distance dependence, we compare experimental force-distance spectra  
199 to analytical and numeric models from the literature. In particular, we combine the models for  
200 the apex contribution of Hudlet et al.[49] with the cone and lever contributions from Colchero  
201 et al.[50,51], respectively. The full equations for the force together with the resulting capacitance  
202 used here are given in the appendix (see eqs. (12) to (17) and Figure 10).

203 In Figure 4 (a) and (b), we compare the respective contributions to the first and second capacitance-  
204 gradients together with the corresponding electrostatic forces during a typical AFM experiment as



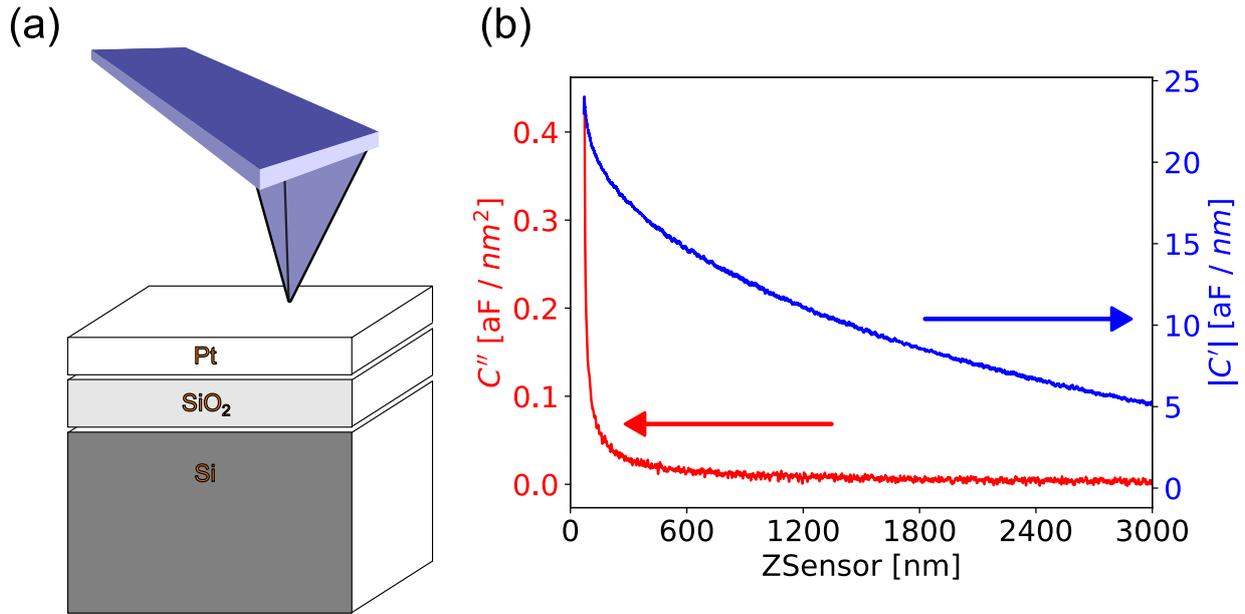
**Figure 4:** Contributions of the respective components to the (a) first numeric derivative  $C'$  and (b) second numeric derivative  $C''$  of the capacitance versus  $z$  distance between tip and sample. Additionally, the respective force (a)  $F(C')$  and (b)  $F(C'')$  was plotted as well against  $z$ . The NuNano SPARK 70 Pt cantilever ( $w = 30 \mu\text{m}$ ,  $l = 225 \mu\text{m}$ ,  $\alpha = 11 \text{ deg}$ ,  $h = 12 \mu\text{m}$ ,  $\theta = 25 \text{ deg}$ ,  $r = 18 \text{ nm}$ ,  $\delta = 3.7 \cdot 10^{-7}$ ) was used for the calculations with an mechanical amplitude of  $A_m = 10 \text{ nm}$ , an excitation voltage of  $V_{AC} = 2 \text{ V}$ , and a total amount of calculated points of 100,000. The blue line marks the apex, the green line the cone, the red line the lever and the black line marks the entire system of the three components in parallel.

205 a function of tip-sample distance  $z$ . For the force calculations, we used eq. (8) together with the pa-  
 206 rameters of a regular EFM cantilever (NuNano SPARK 70 Pt) and an electrical drive of  $V = 2 \text{ V}$   
 207 and a mechanical amplitude of  $A_m = 10 \text{ nm}$ . Comparing the graphs, we can immediately see that  
 208 the total  $C'$  signal retains a significant long-range contribution even at a tip-sample separation of  
 209 3000 nm (Figure 4 (a)). In contrast, the  $C''$  signal shown in drops more rapidly over a short dis-  
 210 tance  $z$  (Figure 4 (b)), indicating a reduced influence of long-ranged contributions to the force sig-  
 211 nals.

212 A measure of how much the signal is disturbed by non-local long-ranged contributions is the ra-

213 tio between the apex contribution to the total signal at a given distance  $z$ . At a typical tip-sample  
214 separation of 10 nm, the apex signal makes up more than 82 % of the complete  $C''$  signal while the  
215 apex contribution on the first capacitance gradient only makes up less than 10 % of the total  $C'$  sig-  
216 nal. In closer proximity of 1 nm distance to the sample, the apex contribution in the  $C''$  signal in-  
217 creases to 99.8 %, whereas the  $C'$  signal still contains a significant amount of non-local signal con-  
218 tributions with 62 % apex vs. 38 % cone and lever signal. Another way to quantify the "locality"  
219 a force signal is to investigate the tip sample separation, where the tip apex contribution surpasses  
220 the lever plus cone contributions within Figure 4. This is true in Figure 4(a) for distances smaller  
221 than  $\approx 3$  nm while in Figure 4 (b) this is the case even for distances smaller than  $\approx 20$  nm. Com-  
222 paring the absolute values of the forces, however, we see that H-SCM yields much weaker forces:  
223 At a tip-sample distance of 10 nm, the AM-based operation leads to a force of  $F_{ES}(C') = 6.7$  nN,  
224 as compared to  $F_{ES}(C'') = 280$  pN for H-SCM. So the resulting electrostatic force and thereby  
225 the expected force is more than a factor of 24 lower for H-SCM. So the improved lateral resolution  
226 comes at the price of a reduced signal-to-noise ratio (SNR).

227 To reproduce these findings experimentally, we performed force-distance spectroscopy on the  
228 etched microcapacitors (Figure 2 in the Methods and Experimental section). The resulting curves  
229 of the  $C'$  and  $C''$  signal qualitatively reproduced the simulation results (Figure 5): Whereas the  $C''$   
230 signal only emerged from the noise at distances of less than 500 nm, the  $C'$  signal shows a mono-  
231 tonic decrease over the full 3  $\mu$ m of vertical travel. Compared to the simulations, the experimental  
232  $C'$  signal shows a slower decrease, indicating a stronger influence by the tip cone. The direct com-  
233 parison of the model and the data of the second and first capacity gradient can be found in Support-  
234 ing Information Files 15 and 16, respectively. These results clearly show that the H-SCM method  
235 produced an electrostatic force signal that is highly local with suppressed stray contributions from  
236 cone and lever.

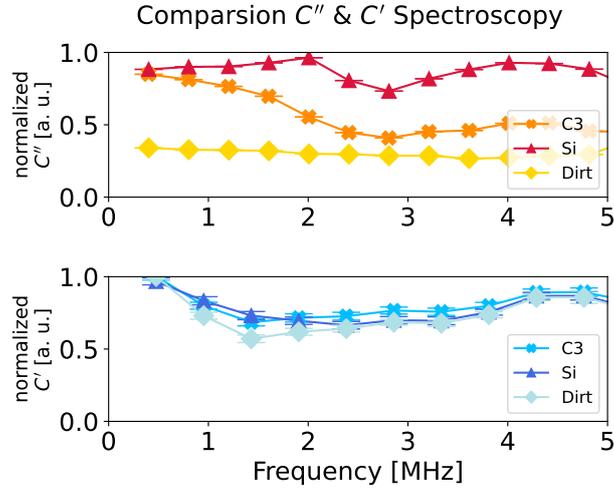


**Figure 5:** Comparison of the  $C''$  and the  $C'$  single force curves (b) of a microcapacitor (a) while doing H-SCM (see eqs. (7b) and (8)) and compared with the detection of  $2\omega$  (see eqs. (3c) and (9)). This was conducted with the NuNano's SPARK 70 Pt cantilever.

## 237 Dielectric spectroscopy

238 The advantage of the multi-frequency excitation approach of H-SCM is that we can choose arbitrary frequencies for the electrostatic excitation. As the tip-sample capacitance is influenced by  
 239 trary frequencies for the electrostatic excitation. As the tip-sample capacitance is influenced by  
 240 the dielectric properties of the material in the tip-sample gap, the frequency-dependent electrostatic force represents the local dielectric function. To demonstrate the feasibility of dielectric  
 241 static force represents the local dielectric function. To demonstrate the feasibility of dielectric  
 242 nano-spectroscopy, we performed H-SCM frequency-spectroscopy in three different locations on  
 243 the microcapacitor sample where we expect a vastly different dielectric response. A first spectrum  
 244 was recorded on one of the microcapacitors (C3, see methods). Then, we measured on the bare Si  
 245 where we expect a fresh native oxide layer of  $\approx 5$  nm thickness (Si). Lastly, we measured on a particle  
 246 of unknown origin (Dirt, visible in Figure 2). The frequency sweeps were performed by keeping  
 247 the tip position and amplitude fixed, varying the two heterodyne excitation frequencies while  
 248 keeping their separation fixed, and recording the resulting excitation amplitude at the second mechanical  
 249 resonance. All spectra were normalized against a reference spectrum recorded on the bare

250 substrate far away from the capacitors to compensate any frequency-response arising from the stray  
 251 capacitance in the signal paths and cantilever.



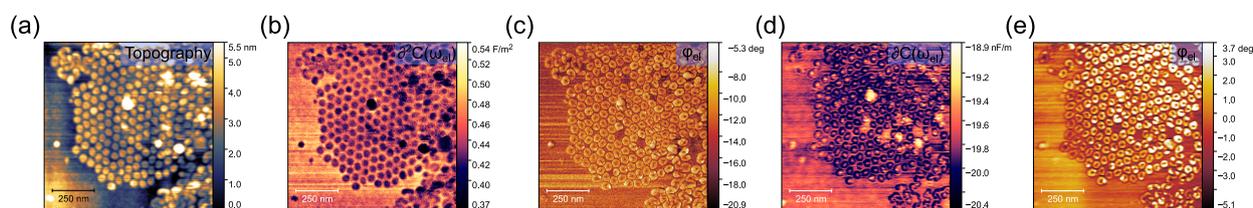
**Figure 6:** Comparison of the normalized  $C''$  (red colors on top) and normalized  $C'$  (blue colors at the bottom) frequency sweep on one of the the capacitors (C3) (cross symbols), the milled Silicon (Si) (triangle symbols), and a measurement on a particle of unknown origin (Dirt) (square symbols). This experiment was conducted with the  $\mu$ masch's HQ:NSC18/Pt Cantilever. The non-normalized data can be seen in Supporting Information Files 2 to 5 and 8 to 11.

252 The electrostatic signal of the capacitor C3 showed a drop at around 2 MHz in Figure 6. When con-  
 253 sidering the capacitance of C3 of  $(183 \pm 1)$  aF and the drop-off frequency  $\omega_d$  of the capacitance  
 254 at 1.7 MHz, we can calculate the resistance  $R$  via the RC time ( $R \cdot C = 1/\omega_d$ ) as  $R \approx 3200$  M $\Omega$ .  
 255 This value is close to the value of 8 G $\Omega$ , taking into account the electrical resistivity of silicon of  
 256  $\rho_{Si} = 2.3 \cdot 10^{12}$  n $\Omega$  m and a thickness of the SiO<sub>2</sub> of 300 nm.  
 257 The  $C''$  signal of the bare Si was stable over the whole range of excitation frequencies and only  
 258 dropped at a much higher frequency around 24 MHz (see Supporting Information File 4). In the  
 259 frequency response of the undefined particle, we found little to no signal response, even at low ex-  
 260 citation frequency. A rise of the signal at around 6 MHz could be observed in all the  $C''$  signals at  
 261 that frequency (see Supporting Information File 2) which we attribute to a capacitive singularity  
 262 in the electrical connection to the sample. We observed a similar behavior in the frequency range  
 263 between 5 and 10 MHz and around 17 MHz. We want to point out that we used standard AFM  
 264 equipment with no special means to control the impedance of the electrical connections. To ob-

265 tain more trustworthy data in the frequency range above 5 MHz will require specialized sample-  
 266 and cantilever holders with coaxial electric connections.

267 To compare these results with the conventional AM-based SCM approach, we repeated the spec-  
 268 troscopy experiments for the  $C'$  signal based on the second term in eq. (7a) (Figure 6, non-  
 269 normalized data in Supporting Information Files 8 to 10). In comparison to the H-SCM data, the  
 270  $C'$  frequency sweep looked very similar on the different structures. We think that this reduction in  
 271 contrast is caused by the stronger influence of long-ranged interactions in the  $C'$  signal, reducing  
 272 the overall impact of the local interaction of the tip apex with the area of interest.

### 273 Imaging $C'$ versus $C''$



**Figure 7:** H-SCM images taken on F14H20 with (a) the topography, (b) the  $C''$  picture at 1.59 and 1.98 MHz, (c) electric phase  $\varphi_{el}$  of the  $C''$  signal at 1.59 and 1.98 MHz, (d) the  $C'$  picture at 235.579 kHz, (e) electric phase  $\varphi_{el}$  of the  $C'$  signal at 235.579 kHz. The full picture can be found in Supporting Information File 14. This was conducted with the  $\mu$ masch's HQ:NSC18/Pt Cantilever.

274 To demonstrate the capabilities of H-SCM as an imaging method, we performed experiments  
 275 on self-assembled nanostructures consisting of the amphiphilic molecule Perfluoroalkyl-Alkane  
 276  $F(CF_2)_{14}(CH_2)_{20}H$  (F14H20) (Figure 7). On the silicon substrate, the F14H20 formed groups of  
 277 spherical particles with a diameter of  $(40 \pm 5)$  nm (Figure 7(a))[52,53]. Simultaneously with the  
 278 topography, we recorded the  $C''$  amplitude and phase at an electrical excitation frequency of 1.59  
 279 and 1.98 MHz. In Figure 7(b) we see a sharp contrast between the F14H20 aggregates and the sil-  
 280 icon substrate. Interestingly, the image of the  $C'$  signal measured at 236 kHz showed a different  
 281 contrast. In particular, both the amplitude and phase contrast of the particles changes from left to  
 282 right, indicating long ranged background signal.

## 283 **Conclusion**

284 In this paper, we have presented a novel method for high-resolution nanoscale capacitance charac-  
285 terization based on multi-frequency electrostatics. The key advantage of the multi-frequency ap-  
286 proach of H-SCM is that it allows for operation at almost arbitrary frequencies, enabling the mea-  
287 surement of the local dielectric function over a wide range of frequencies. In comparison to exist-  
288 ing SCM operation modes, H-SCM leads to a significant reduction of signal background, which re-  
289 sults in higher locality of the measurements with less cross-talk. This is due to the fact that the sec-  
290 ond capacitance gradient is less affected by long-range interactions, such as those from the tip cone  
291 and lever. We demonstrate the reliable operation using standard AFM equipment together with an  
292 external LIA up to a frequency of 5 MHz. At higher frequencies (up to 50 MHz in our case), the  
293 signals were dominated by impedance effects from the signal connections. Thus, to move towards  
294 reliable measurements at higher frequencies, specialized HF-equipment with coaxial signal connec-  
295 tions will be required.

296 Our analytical simulations of the distance-dependent tip-sample capacitance showed that current  
297 models are not able to fully simulate the experimental data. Thus, to enable quantitative measure-  
298 ments of the tip-sample capacitance, further measures such as improved tip-sample models or full  
299 numerical simulations will be required. This will pave the way for quantitative studies of dielectric  
300 effects in nanoscale systems in materials science, biology, and nanotechnology.

## 301 **Experimental (optional)**

### 302 **Polymer blend samples**

303 We used the Perfluoroalkyl-Alkane  $F(CF_2)_{14}(CH_2)_{20}H$  (F14H20) samples that we bought from  
304 SPM Labs LLC.

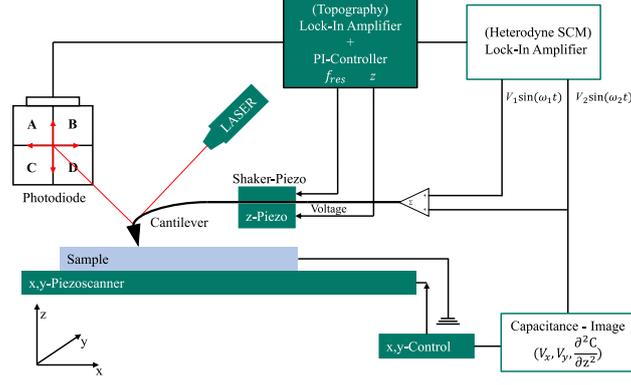
## 305 **Microcapacitors**

306 The Si wafers "CZ" were bought from "Si-Mat" with a diameter of 150 mm, a surface orientation  
307  $\langle 100 \rangle$ , a thickness of  $(675 \pm 20) \mu\text{m}$ , a resistivity of 1.5 - 4.0  $\Omega \text{cm}$ , and with a p-type doping  
308 with B-atoms. These wafers were thermally oxidised with 300 nm  $\text{SiO}_2$ . To sputter Pt on top of the  
309 silicon wafer the Pt sputter machine Compact Coating Unit (CCU) 010/LV with the sputter head  
310 SP010 was used to sputter 14 nm on top of the wafer. The microcapacitors were then milled out of  
311 the surface by a FIB from FEI Nova600 Nanolab with a dual-beam  $\text{Ga}^+$  ion beam.

## 312 **Heterodyne Scanning Capacitance Microscopy (H-SCM) Measurements**

313 H-SCM was measured on an Oxford Instruments/Asylum Research MFP-3D Infinity AFM in a  
314 nitrogen glovebox (level of humidity below 0.3 %, level of oxygen below 0.1 %) for all experi-  
315 ments. The typical resonance frequency of the Pt/Ir coated conductive cantilevers (NuNano model:  
316 SPARK-150Pt;  $\mu\text{masch}$  model: HQ:NSC18/Pt) was  $\approx 75 \text{ kHz}$ , spring constant of 2 to 3  $\frac{\text{N}}{\text{m}}$ , a tip  
317 radius of 18 nm and a tip height of 10 to 18  $\mu\text{m}$ . The topography feedback was performed with am-  
318 plitude modulation (AM) on the first eigenmode  $\omega_{m,1}$  and the oscillation amplitude was kept to ap-  
319 proximately  $\approx 70 - 90 \text{ nm}$  for all measurements. The force spectroscopy measurements were done  
320 with a  $z$ -Rate of 0.2 Hz and a force distance of 8  $\mu\text{m}$  for all samples.

321 We used a Zurich Instruments HF2 lock-in amplifier for all experiments including to perform the  
322 H-SCM measurements. The electric drive amplitude of the  $V_{AC,1} = V_{AC,2}$  signal varied between 3  
323 and 5 V depending on the obtained signal from the sample. We grounded the sample via the sam-  
324 ple holder with an external wire to ground level of the Zürich Lock-In Amplifier. The applied  $V_{AC}$   
325 was applied to the tip directly while the AFM head connections were switched off. The setup of  
326 the AFM is shown in Figure 8. The electrical connection from the LIA to the cantilever with the  
327 two excitation voltages was realized by using a direct cable connection. The sample was always  
328 grounded to the ground level of the LIA. A scheme can be seen in Figure 8.



**Figure 8:** Schematic setup of H-SCM. Additionally to a regular AFM, two different voltages are applied to the cantilever with different frequencies respectively.

### 329 Focused ion beam (FIB)

330 FIB of the cantilever was conducted using a LEO Gemini instrument from Zeiss. It was used with  
 331 an acceleration voltage of 3 kV.

## 332 Appendix

### 333 Equations to calculate the $C''$ and $C'$ Signal from the voltages

334 Equation (10) shows the detailed expression shown in eq. (8). Within the detected amplitude from  
 335 the LIA  $A_{\text{det}}$  is the Voltage from the LIA ( $V_{\text{H-SCM}}$ ) and the amplification factor  $\Xi_{\text{amp,d2C}}$  of this  
 336 voltage from the LIA in H-SCM mode. The frequency dependent spring constant  $k(\omega)$  in eq. (8)  
 337 contains the inverse optical lever sensitivity (InvOLS) of the second harmonic ( $\text{InvOLS}_2$ ) and  
 338 the spring constant of the second resonance ( $k_2$ ) shown in eq. (10). It is important to note that the  
 339 InvOLS and the spring constant on the seconds resonance is not the same as measured on the first  
 340 resonance by the method of Sader et al.[54]. It is rather necessary to calculate the properties of the  
 341 cantilever for the respective eigenmodes[55].

$$342 \quad \frac{\partial^2 C}{\partial z^2}(\omega) = C''(\omega) = \frac{16 \cdot V_{\text{H-SCM}}(\omega) \cdot \text{InvOLS}_2(\omega) \cdot k_2(\omega)}{A_m \cdot V_{\text{AC}}^2 \cdot \Xi_{\text{amp,d2C}}} \quad (10)$$

Equation (11) shows the detailed expression shown in eq. (9). In the expression  $A_{\text{det}}$  is the detected voltage from the LIA ( $V_{\text{SF-EFM}}$ ) and again an amplification factor  $\Xi_{\text{amp,dC}}$  of the signal captured with the LIA with the SF-EFM mode. The frequency-dependent spring constant  $k(\omega)$  is the same as above and consists of  $\text{InvOLS}_2$  and  $k_2$ .

$$\frac{\partial C}{\partial z}(\omega) = C'(\omega) = \frac{4 \cdot V_{\text{SF-EFM}}(\omega) \cdot \text{InvOLS}_2(\omega) \cdot k_2(\omega)}{V_{\text{AC}}^2 \cdot \Xi_{\text{amp,dC}}} \quad (11)$$

### Full double excitation force equations

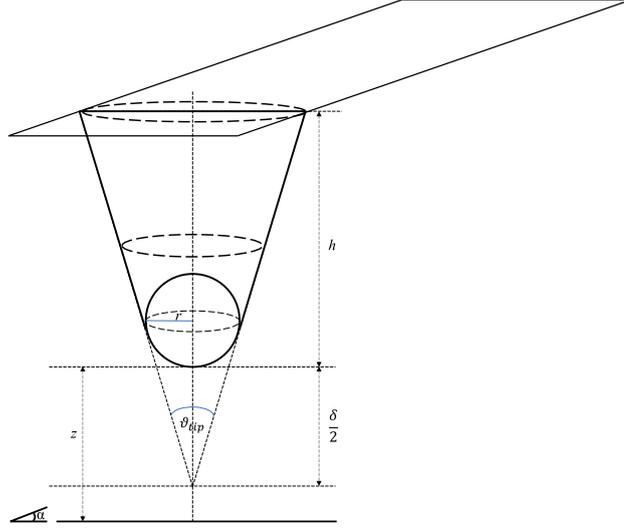
Full overview of the electric amplitude contributions at various frequencies while activating the heterodyne Scanning Capacitance Microscopy (H-SCM) mode. For simplicity, we will use the following substitutions  $\omega_m^1 t = O$ ,  $\omega_e t = E$ ,  $\omega_{\text{mod}} t = M$ ,  $V_{\text{CPD}} - V_{\text{DC}} = \Delta$ ,  $\hat{V}_{\text{AC}} = V$ , and  $A_m^1 = A$ . In Table 1 is the overview of the force components at various frequencies for the resulting static,  $\omega$  and  $2\omega$  force components acting on the cantilever.

**Table 1:** Overview of the components of the multi-frequency electrostatic force microscopy.

| Frequency          | Amplitude                   |
|--------------------|-----------------------------|
| DC                 | $1/2C'[\Delta^2 + U^2/4]$   |
| $2M$               | $1/8C'U^2$                  |
| $O$                | $1/2C''A[\Delta^2 + U^2/4]$ |
| $O \pm 2M$         | $1/16C''AU^2$               |
| $E \pm M$          | $1/2C'U \Delta$             |
| $O \pm (E \pm M)$  | $1/4C''AU\Delta$            |
| $2E$               | $1/8C'U^2$                  |
| $2(M \pm E)$       | $1/16C'U^2$                 |
| $O \pm 2E$         | $1/16C''AU^2$               |
| $O \pm 2(E \pm M)$ | $1/32C''AU^2$               |

### Tip-sample capacity model

We used the model of Hudlet et al.[49] for the tip apex and in addition used the sum of cone and lever distribution of Colchero et al.[50,51]. The cantilever can be modeled as a tilted plate capacitor with a truncated cone at the end of the cantilever and with a sharp round tip apex at the end of the tip cone. This is shown schematically in Figure 9.



**Figure 9:** Auxiliary sketch of the capacitance model of the truncated cone with spherical apex. Here  $h$  is the height of the tip,  $r$  is the radius of the sphere,  $\vartheta_{\text{tip}}$  is the opening angle of the tip,  $\delta$  is the truncated part of the cone and  $z$  is the distance between sample and tip apex in respect to the surface normal of the sample.  $\alpha$  is the angle between the surface and the lever of the cantilever.

359 In this case the electrostatic force for the lever is given by eq. (12).

$$360 \quad F_{\text{lever}}(z) = \frac{2 \tan^2(\frac{\alpha}{2})}{\alpha^2} \varepsilon_0 V_{\text{tip-sample}}^2 \frac{l w}{h^2} \frac{1}{\left[ \left(1 + \frac{z}{h}\right) \cdot \left(1 + \frac{z+2l \tan(\frac{\alpha}{2})}{h}\right) \right]} [50,51]. \quad (12)$$

361 Integration due to eq. (1) yields

$$362 \quad C_{\text{lever}}(z) = \frac{2 \tan^2(\frac{\alpha}{2})}{\alpha^2} \varepsilon_0 V_{\text{tip-sample}}^2 \frac{l w}{h^2} \frac{h^2 \cot(\frac{\alpha}{2}) (\ln(h+z) - \ln(\cos(\frac{\alpha}{2})(h+z) + 2l \sin(\frac{\alpha}{2})))}{2l}, \quad (13)$$

363 where  $\varepsilon_0$  is the dielectric constant of the vacuum. The dimensions of the lever are given by its

364 width  $w$ , its length  $l$  and the height of the tip cone  $h$ . The lever is tilted by the angle  $\alpha = \vartheta_{\text{lever}}$ .

365 The tip cone can be approximated by a truncated cone (Figure 9). The electrostatic force as a func-

tion of distance between tip cone and sample is given by eq. (14).

$$F_{\text{cone}}(z) = \frac{4 \pi}{(\pi - \vartheta_{\text{tip}})^2} \varepsilon_0 V_{\text{tip-sample}}^2 \left[ \ln \left( \frac{z - \frac{\delta}{2} + h}{z + \frac{\delta}{2}} \right) - \sin \left( \frac{\vartheta_{\text{tip}}}{2} \right) \frac{h - \delta}{z - \frac{\delta}{2} + h} \cdot \frac{z - \frac{\delta}{2}}{z + \frac{\delta}{2}} \right] [50,51] \quad (14)$$

with the open angle of the tip cone ( $\vartheta_{\text{tip}}$ ), and the height of the truncated part of the cone ( $\delta = r/\tan^2(\vartheta_{\text{tip}}/2)$ )[50,51]. Integration of this equation to obtain the capacitance yields

$$C_{\text{cone}}(z) = 2 \frac{4\pi\varepsilon_0}{(\vartheta_{\text{tip}} - \pi)^2} \left[ \sin \left( \frac{\vartheta_{\text{tip}}}{2} \right) (h \ln(2f_1) - \delta \ln(f_2)) + f_1 \ln \left( \frac{f_2}{2f_1} \right) + (\delta - h) \ln(f_2) \right], \quad (15)$$

where  $f_1 = z - \frac{\delta}{2} + h$  and  $f_2 = 2z + \delta$ .

The tip apex is approximated as a sphere over an infinite surface (Figure 9). The corresponding electrostatic force between a tip apex and the surface is given in eq. (16).

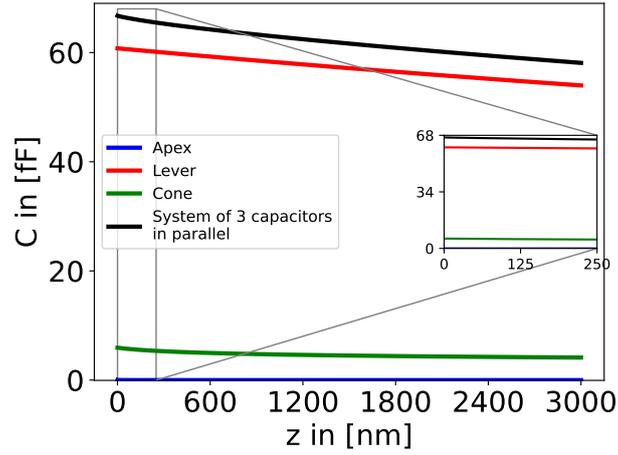
$$F_{\text{apex}}(z) = \pi \varepsilon_0 r^2 V_{\text{tip-sample}}^2 \left[ \frac{1 - \sin \left( \frac{\vartheta_{\text{tip}}}{2} \right)}{z \left( z + r \left( 1 - \sin \left( \frac{\vartheta_{\text{tip}}}{2} \right) \right) \right)} \right] [49]. \quad (16)$$

Hence, the capacitance is given by

$$C_{\text{apex}}(z) = 2\pi\varepsilon_0 r \ln \left( \frac{z + r \left( 1 - \sin \left( \frac{\vartheta_{\text{tip}}}{2} \right) \right)}{z} \right). \quad (17)$$

When you plot the capacity of the cantilever against the distance between the tip and the sample  $z$  you get Figure 10. The parameter were taken from the Website of the producer of the NuNano SPARK 70 Pt cantilever:  $w = 30 \mu\text{m}$ ,  $l = 225 \mu\text{m}$ ,  $\alpha = 11 \text{ deg}$ ,  $h = 12 \mu\text{m}$ ,  $\vartheta_{\text{cone}} = 25 \text{ deg}$ ,  $r = 18 \text{ nm}$ , and  $V_{\text{AC}} = 2 \text{ V}$ .

In order to get the first  $C'$  and second capacity gradient  $C''$  of the relevant parts of the cantilever,



**Figure 10:** Contributions of the respective components to the numeric capacity  $C$  versus  $z$  distance between tip and sample. The properties of the NuNano SPARK 70 Pt cantilever ( $w = 30 \mu\text{m}$ ,  $l = 225 \mu\text{m}$ ,  $\alpha = 11 \text{ deg}$ ,  $h = 12 \mu\text{m}$ ,  $\theta = 25 \text{ deg}$ ,  $r = 18 \text{ nm}$ ,  $\delta = 3.7 \cdot 10^{-7}$ ) with an mechanical amplitude of  $A_m = 10 \text{ nm}$ , an excitation voltage of  $V_{AC} = 2 \text{ V}$ , and a total amount of calculated points of 100,000, was used for the calculations. The blue line marks the apex, the green line the cone, the red line the lever and the black line marks the entire system of the three components in parallel.

382 we used the onward and backward differentiation seen in eq. (18) and the central differential quo-  
 383 tient of the second order seen in eq. (19) is used, respectively.

$$384 \quad f'(x) = \frac{f(x+h) - f(x-h)}{2h} \quad (18)$$

385 The step size was chosen to be  $1 \cdot 10^{-10} \text{ m}$  with a total amount of 1,000,000 steps.

$$386 \quad f''(x) = \frac{f(x+h) - 2f(x) + f(x-h)}{h^2} \quad (19)$$

387 Again, the step size of this was chosen to be the same as for the first derivative.

388 The model of the first and second capacity gradient can be found in Figure 4 (a) and (b), respec-  
 389 tively.

## 390 **Supporting Information**

391 Supporting information features a comparison of the working principles of H-KPFM and H-SCM,  
392 all the raw and normalized data of the H-SCM frequency spectroscopy, the full comparison of the  
393 H-SCM, SF-EFM, and H-KPFM images on the F14H20 structures, and finally a comparison of the  
394 model data and the measured data on the microcapacitors.

395 Supporting Information File 1:

396 File Name: H-KPFM\_and\_H-SCM\_scheme\_working\_principle\_two.pdf

397 File Format: PDF

398 Title: Schematic comparison of the excitation and detection frequencies in H-KPFM and H-SCM.

399 The lower part shows the transfer function of the cantilever, where the amplitude is plotted vs the  
400 logarithmic angular frequency. The upper part shows the excitation frequencies ( $\downarrow$ ) and the detec-  
401 tion frequencies ( $\uparrow$ ) of the applied frequencies. The red arrow corresponds to topography- and the  
402 blue arrow to the electrical signal[18,26,56].

403 Supporting Information File 2:

404 File Name: Freq\_Sweep\_Comparison\_d2C\_vs\_Frequency.pdf

405 File Format: PDF

406 Title: Non-normalized data of the comparison of the  $C''$  frequency sweep shown in Figure 6 on  
407 the four spots while in H-SCM (see eq. (8)). This was conducted with the  $\mu$ masch's HQ:NSC18/Pt  
408 cantilever.

409 Supporting Information File 3:

410 File Name: Freq\_Sweep\_Comparison\_d2C\_vs\_Frequency\_zoom.pdf

411 File Format: PDF

412 Title: Zoom of the non-normalized data from the comparison of the  $C''$  frequency sweep shown  
413 in Figure 6 on the four spots while in H-SCM (see eq. (8)). This was conducted with the  $\mu$ masch's  
414 HQ:NSC18/Pt cantilever.

415 Supporting Information File 4:

416 File Name: Freq\_Sweep\_Comparison\_d2C\_vs\_Frequency\_normed.pdf

417 File Format: PDF

418 Title: Normalized data of the  $C''$  frequency sweep shown in Figure 6 on the three spots while in  
419 H-SCM (see eq. (8)). This was conducted with the  $\mu$ masch's HQ:NSC18/Pt cantilever.

420 Supporting Information File 5:

421 File Name: Freq\_Sweep\_Comparison\_d2C\_vs\_Frequency\_normed\_zoom.pdf

422 File Format: PDF

423 Title: Zoomed and normalized data of the  $C''$  frequency sweep shown in Figure 6 on the three  
424 spots while in H-SCM (see eq. (8)). This was conducted with the  $\mu$ masch's HQ:NSC18/Pt can-  
425 tilever.

426 Supporting Information File 6:

427 File Name: Freq\_Sweep\_Comparison\_elec\_Phase\_d2C\_vs\_Frequency.pdf

428 File Format: PDF

429 Title: Non-normalized data of the phase signal  $\varphi$  spectra of the comparison from the  $C''$  frequency  
430 sweep shown in Figure 6 on the four spots while in H-SCM (see eq. (8)). This was conducted with  
431 the  $\mu$ masch's HQ:NSC18/Pt cantilever.

432 Supporting Information File 7:

433 File Name: Freq\_Sweep\_Comparison\_elec\_Phase\_d2C\_vs\_Frequency\_zoom.pdf

434 File Format: PDF

435 Title: Zoom of the non-normalized data of the phase signal  $\varphi$  spectra of the comparison from the  
436  $C''$  frequency sweep shown in Figure 6 on the four spots while in H-SCM (see eq. (8)). This was  
437 conducted with the  $\mu$ masch's HQ:NSC18/Pt cantilever.

438 Supporting Information File 8:

439 File Name: Freq\_Sweep\_Comparison\_dC\_vs\_Frequency.pdf

440 File Format: PDF

441 Title: Non-normalized data of the comparison of the  $C'$  frequency sweep shown in Figure 6

442 on the four spots while in SF-EFM mode (see eq. (9)). This was conducted with the  $\mu$ masch's  
443 HQ:NSC18/Pt cantilever.

444 Supporting Information File 9:

445 File Name: Freq\_Sweep\_Comparison\_dC\_vs\_Frequency\_zoom.pdf

446 File Format: PDF

447 Title: Zoom of the non-normalized data of the comparison of the  $C'$  frequency sweep shown in  
448 Figure 6 on the four spots while in SF-EFM mode (see eq. (9)). This was conducted with the  
449  $\mu$ masch's HQ:NSC18/Pt cantilever.

450 Supporting Information File 10:

451 File Name: Freq\_Sweep\_Comparison\_dC\_vs\_Frequency\_normed.pdf

452 File Format: PDF

453 Title: Normalized data of the  $C'$  frequency sweep shown in Figure 6 on the three spots while in  
454 SF-EFM mode (see eq. (9)). This was conducted with the  $\mu$ masch's HQ:NSC18/Pt cantilever.

455 Supporting Information File 11:

456 File Name: Freq\_Sweep\_Comparison\_dC\_vs\_Frequency\_normed\_zoom.pdf

457 File Format: PDF

458 Title: Zoomed and normalized data of the  $C'$  frequency sweep shown in Figure 6 on the three spots  
459 while in SF-EFM mode (see eq. (9)). This was conducted with the  $\mu$ masch's HQ:NSC18/Pt can-  
460 tilever.

461 Supporting Information File 12:

462 File Name: Freq\_Sweep\_Comparison\_elec\_Phase\_dC\_vs\_Frequency.pdf

463 File Format: PDF

464 Title: Non-normalized data of the phase signal  $\varphi$  spectra of the comparison of the  $C'$  frequency  
465 sweep shown in Figure 6 on the four spots while in SF-EFM mode (see eq. (9)). This was con-  
466 ducted with the  $\mu$ masch's HQ:NSC18/Pt cantilever.

467 Supporting Information File 13:

468 File Name: Freq\_Sweep\_Comparison\_elec\_Phase\_dC\_vs\_Frequency\_zoom.pdf

469 File Format: PDF

470 Title: Zoom of the non-normalized data of the phase signal  $\varphi$  spectra of the comparison of the  $C'$   
471 frequency sweep shown in Figure 6 on the four spots while in SF-EFM mode (see eq. (9)). This  
472 was conducted with the  $\mu$ masch's HQ:NSC18/Pt cantilever.

473 Supporting Information File 14:

474 File Name: Zeichnung\_F14H20\_Comparison\_dC\_d2C\_SI.pdf

475 File Format: PDF

476 Title: Full version of the H-SCM pictures given in Figure 7. H-SCM pictures made on F14H20  
477 with (a) the topography, (b) the  $C''$  picture at 1.59 and 1.98 MHz, (c) electric phase  $\varphi_{el}$  of the  $C''$   
478 signal at 1.59 and 1.98 MHz, (d) the  $C'$  picture at 235.579 kHz, (e) electric phase  $\varphi_{el}$  of the  $C'$  sig-  
479 nal at 235.579 kHz, (f) the CPD picture, (g) the  $C''$  picture at 15.88 and 16.28 MHz, (h) electric  
480 phase  $\varphi_{el}$  of the  $C''$  signal at 15.88 and 16.28 MHz, (i) the picture of the mechanical amplitude at  
481 the resonance frequency of 74.580 kHz, and (j) the picture of the mechanical phase at the reso-  
482 nance frequency of 74.580 kHz. This was conducted with the  $\mu$ masch's HQ:NSC18/Pt Cantilever.

483 Supporting Information File 15:

484 File Name: Comparison\_Model\_&\_Data\_d2C\_vs\_ZSensor\_data\_model\_2\_500.pdf

485 File Format: PDF

486 Title: A comparison of the measured  $C''$  values on various capacitors, as shown in Figure 2, is pre-  
487 sented. The measurements, performed using the NuNano SPARK 70 Pt cantilever (solid lines), are  
488 contrasted with the theoretical contributions of the respective components to the first numerical  
489 derivative  $C'$  of the capacitance (dotted lines) as a function of the tip-to-sample distance,  $z$ . For  
490 the theoretical calculations, the properties of the NuNano SPARK 70 Pt cantilever ( $w = 30 \mu\text{m}$ ,  
491  $l = 225 \mu\text{m}$ ,  $\alpha = 11 \text{ deg}$ ,  $h = 12 \mu\text{m}$ ,  $\theta = 25 \text{ deg}$ ,  $r = 18 \text{ nm}$ ,  $\delta = 3.7 \cdot 10^{-7}$ ) with an mechanical  
492 amplitude of  $A_m = 10 \text{ nm}$ , an excitation voltage of  $V_{AC} = 2 \text{ V}$ , and a total amount of calculated  
493 points of 100,000, was used for these.

494 Supporting Information File 16:

495 File Name: Comparison\_Model\_&\_Data\_dC\_vs\_ZSensor\_data\_model\_2\_500.pdf

496 File Format: PDF

497 Title: A comparison of the measured  $C'$  values on various capacitors, as shown in Figure 2, is pre-  
498 sented. The measurements, performed using the NuNano SPARK 70 Pt cantilever (solid lines), are  
499 contrasted with the theoretical contributions of the respective components to the first numerical  
500 derivative  $C'$  of the capacitance (dotted lines) as a function of the tip-to-sample distance,  $z$ . For  
501 the theoretical calculations, the properties of the NuNano SPARK 70 Pt cantilever ( $w = 30 \mu\text{m}$ ,  
502  $l = 225 \mu\text{m}$ ,  $\alpha = 11 \text{ deg}$ ,  $h = 12 \mu\text{m}$ ,  $\theta = 25 \text{ deg}$ ,  $r = 18 \text{ nm}$ ,  $\delta = 3.7 \cdot 10^{-7}$ ) with an mechanical  
503 amplitude of  $A_m = 10 \text{ nm}$ , an excitation voltage of  $V_{AC} = 2 \text{ V}$ , and a total amount of calculated  
504 points of 100,000, was used for these.

## 505 Acknowledgements

506 This work is based on the preliminary results previously reported in the Bachelor Thesis in Physics  
507 named “Tip-sample capacitance in electrostatic force microscopy” by Peter Gregor Reichel (2021,  
508 Johannes Gutenberg-Universität Mainz). We acknowledge Dr. Pravash Bista giving helpful ideas  
509 in the process of the research. Also, we acknowledge Maren Müller, our technician, for the help to  
510 prepare the capacitors.

## 511 References

- 512 1. Elayarani, P.; Sumathi, T.; Sivakumar, G.; Pragadeswaran, S.; Suthakaran, S.; Sathiyamurthy, S.; Seshadhri, J.; Ayyar, M.; Arularasu, M. V. *Zeitschrift für Phys. Chemie* **2024**, *238*  
513 (6), 1–24. doi:10.1515/zpch-2023-0531.
- 515 2. Majid, F.; Bashir, M.; Bibi, I.; Ayub, M.; Khan, B. S.; Somaily, H. H.; Al-Mijalli, S. H.;  
516 Nazir, A.; Iqbal, S.; Iqbal, M. *Zeitschrift für Phys. Chemie* **2023**, *237* (9), 1345–1360. doi:  
517 10.1515/zpch-2022-0097.

- 518 3. Iqbal, Z.; Imran, M.; Latif, S.; Nazir, A.; Ibrahim, S. M.; Ahmad, I.; Iqbal, M.; Iqbal, S.  
519 *Zeitschrift für Phys. Chemie* **2023**, *237* (8), 1139–1152. doi:10.1515/zpch-2022-0113.
- 520 4. Ali, F.; Safdar, A.; Younas, U.; Sillanpaa, M.; Pervaiz, M.; Nazir, A.; Naeem, M.; Iqbal, M.;  
521 Al-Kahtani, A. A.; Tighezza, A. M. *Zeitschrift für Phys. Chemie* **2023**, *237* (4-5), 599–616.  
522 doi:10.1515/zpch-2022-0098.
- 523 5. Nazir, A.; Alam, S.; Alwadai, N.; Abbas, M.; Bibi, I.; Ali, A.; Ahmad, N.; Al Huwayz, M.;  
524 Iqbal, M. *Zeitschrift für Phys. Chemie* **2023**, *237* (11), 1733–1751. doi:10.1515/  
525 zpch-2023-0331.
- 526 6. Arif, H.; Yasir, M.; Ali, F.; Nazir, A.; Ali, A.; Al Huwayz, M.; Alwadai, N.; Iqbal, M.  
527 *Zeitschrift für Phys. Chemie* **2023**, *237* (6), 689–705. doi:10.1515/zpch-2023-0224.
- 528 7. Sahoo, K. K.; Pradhan, D.; Ghosh, S. P.; Gartia, A.; Kar, J. P. *Phys. Scr.* **2024**, *99* (2), 025934.  
529 doi:10.1088/1402-4896/ad196b.
- 530 8. Zhu, C.; Liu, A.; Liu, G.; Jiang, G.; Meng, Y.; Fortunato, E.; Martins, R.; Shan, F. *J. Mater.*  
531 *Chem. C* **2016**, *4* (45), 10715–10721. doi:10.1039/C6TC02607A.
- 532 9. Paily, R.; DasGupta, A.; DasGupta, N.; Bhattacharya, P.; Misra, P.; Ganguli, T.;  
533 Kukreja, L. M.; Balamurugan, A.; Rajagopalan, S.; Tyagi, A. *Appl. Surf. Sci.* **2002**, *187* (3-  
534 4), 297–304. doi:10.1016/S0169-4332(01)01040-6.
- 535 10. Robertson, J. *Reports Prog. Phys.* **2006**, *69* (2), 327–396. doi:10.1088/0034-4885/69/2/R02.
- 536 11. Werner, F.; Babbe, F.; Elanzeery, H.; Siebentritt, S. *Prog. Photovoltaics Res. Appl.* **2019**, *27*  
537 (11), 1045–1058. doi:10.1002/pip.3196.
- 538 12. *Impedance Spectroscopy*; Barsoukov, E., Macdonald, J. R., Eds.; Wiley, 2005; doi:10.1002/  
539 0471716243.
- 540 13. Shi, N.; Ramprasad, R. *J. Comput. Mater. Des.* **2007**, *14* (1), 133–139. doi:10.1007/  
541 s10820-006-9034-9.

- 542 14. Pang, H.-S.; Xu, H.; Tang, C.; Meng, L.-K.; Ding, Y.; Xiao, J.; Liu, R.-L.; Pang, Z.-Q.;  
543 Huang, W. *Org. Electron.* **2019**, *65* (September 2018), 275–299. doi:10.1016/j.orgel.2018.  
544 09.025.
- 545 15. Yalcinkaya, Y.; Rohrbeck, P. N.; Schütz, E. R.; Fakharuddin, A.; Schmidt-Mende, L.; We-  
546 ber, S. A. *Adv. Opt. Mater.* **2024**, *12* (8), 1–12. doi:10.1002/adom.202301318.
- 547 16. Binnig, G.; Rohrer, H.; Gerber, C.; Weibel, E. *Phys. Rev. Lett.* **1982**, *49* (1), 57–61. doi:10.  
548 1103/PhysRevLett.49.57.
- 549 17. Binnig, G.; Quate, C. F.; Gerber, C. *Phys. Rev. Lett.* **1986**, *56* (9), 930–933. doi:10.1103/  
550 PhysRevLett.56.930.
- 551 18. Axt, A.; Hermes, I. M.; Bergmann, V. W.; Tausendpfund, N.; Weber, S. A. L. *Beilstein J. Nan-*  
552 *otechnol.* **2018**, *9* (1), 1809–1819. doi:10.3762/bjnano.9.172.
- 553 19. Hermes, I. M.; Best, A.; Winkelmann, L.; Mars, J.; Vorpahl, S. M.; Mezger, M.; Collins, L.;  
554 Butt, H.-J.; Ginger, D. S.; Koynov, K.; Weber, S. A. L. *Energy Environ. Sci.* **2020**, *13* (11),  
555 4168–4177. doi:10.1039/d0ee01016b.
- 556 20. Rohrbeck, P. N. Untersuchung von lokalen Aufladungsdynamiken an Perowskit-Korngrenzen  
557 mit zeitaufgelösten KPFM Methoden. Master thesis, Johannes Gutenberg University Mainz,  
558 2021.
- 559 21. Bergmann, V. W.; Weber, S. A. L.; Javier Ramos, F.; Nazeeruddin, M. K.; Grätzel, M.; Li, D.;  
560 Domanski, A. L.; Lieberwirth, I.; Ahmad, S.; Berger, R. *Nat. Commun.* **2014**, *5* (1), 5001.  
561 doi:10.1038/ncomms6001.
- 562 22. Weber, S. A. L.; Hermes, I. M.; Turren-Cruz, S.-H.; Gort, C.; Bergmann, V. W.; Gilson, L.;  
563 Hagfeldt, A.; Graetzel, M.; Tress, W.; Berger, R. *Energy Environ. Sci.* **2018**, *11* (9),  
564 2404–2413. doi:10.1039/C8EE01447G.

- 565 23. Zhu, C.; Fuchs, T.; Weber, S. A. L.; Richter, F. H.; Glasser, G.; Weber, F.; Butt, H.-J.;  
566 Janek, J.; Berger, R. *Nat. Commun.* **2023**, *14* (1), 1300. doi:10.1038/s41467-023-36792-7.
- 567 24. Martin, Y.; Abraham, D. W.; Wickramasinghe, H. K. *Appl. Phys. Lett.* **1988**, *52* (13),  
568 1103–1105. doi:10.1063/1.99224.
- 569 25. Abraham, D. W. *J. Vac. Sci. Technol. B Microelectron. Nanom. Struct.* **1991**, *9* (2), 703. doi:  
570 10.1116/1.585536.
- 571 26. Garrett, J. L.; Munday, J. N. *Nanotechnology* **2016**, *27* (24), 14. doi:10.1088/0957-4484/27/  
572 24/245705.
- 573 27. Eaton, P.; West, P. *MRS Bull.*; Oxford University Press, 2010; Vol. 39, pp 379–379. doi:10.  
574 1093/acprof:oso/9780199570454.001.0001.
- 575 28. Biberger, R.; Benstetter, G.; Goebel, H.; Hofer, A. *Microelectron. Reliab.* **2010**, *50* (9-11),  
576 1511–1513. doi:10.1016/j.microrel.2010.07.052.
- 577 29. Lee, D. T.; Pelz, J. P.; Bhushan, B. *Nanotechnology* **2006**, *17* (5), 1484–1491. doi:10.1088/  
578 0957-4484/17/5/054.
- 579 30. Gomila, G.; Toset, J.; Fumagalli, L. *J. Appl. Phys.* **2008**, *104* (2), 024315. doi:10.1063/1.  
580 2957069.
- 581 31. Fumagalli, L.; Ferrari, G.; Sampietro, M.; Gomila, G. *Appl. Phys. Lett.* **2007**, *91* (24), 243110.  
582 doi:10.1063/1.2821119.
- 583 32. Lee, D. T.; Pelz, J. P.; Bhushan, B. *Rev. Sci. Instrum.* **2002**, *73* (10), 3525–3533. doi:10.1063/  
584 1.1505655.
- 585 33. Kobayashi, K.; Yamada, H.; Matsushige, K. *Appl. Phys. Lett.* **2002**, *81* (14), 2629–2631. doi:  
586 10.1063/1.1510582.

- 587 34. Kimura, K.; Kobayashi, K.; Yamada, H.; Matsushige, K. *Appl. Surf. Sci.* **2003**, *210* (1-2),  
588 93–98. doi:10.1016/S0169-4332(02)01486-1.
- 589 35. Gil, A.; Colchero, J.; Gomez Herrero, J.; Bar, A. M. *Nanotechnology* **2003**, *14* (2), 332–340.  
590 doi:10.1088/0957-4484/14/2/345.
- 591 36. Platz, D.; Tholén, E. A.; Pesen, D.; Haviland, D. B. *Appl. Phys. Lett.* **2008**, *92* (15), 153106.  
592 doi:10.1063/1.2909569.
- 593 37. Borgani, R.; Forchheimer, D.; Bergqvist, J.; Thorén, P.-A.; Inganäs, O.; Haviland, D. B. *Appl.*  
594 *Phys. Lett.* **2014**, *105* (14), 143113. doi:10.1063/1.4897966.
- 595 38. Garrett, J. L.; Leite, M. S.; Munday, J. N. *ACS Appl. Mater. Interfaces* **2018**, *10* (34),  
596 28850–28859. doi:10.1021/acsami.8b08097.
- 597 39. Dobryden, I.; Borgani, R.; Rigoni, F.; Ghamgosar, P.; Concina, I.; Almqvist, N.; Vomiero, A.  
598 *Nanoscale Adv.* **2021**, *3* (15), 4388–4394. doi:10.1039/D1NA00319D.
- 599 40. Crider, P. S.; Majewski, M. R.; Zhang, J.; Oukris, H.; Israeloff, N. E. *Appl. Phys. Lett.* **2007**,  
600 *91* (1), 2–5. doi:10.1063/1.2753539.
- 601 41. Labardi, M.; Prevosto, D.; Nguyen, K. H.; Capaccioli, S.; Lucchesi, M.; Rolla, P. *J. Vac.*  
602 *Sci. Technol. B, Nanotechnol. Microelectron. Mater. Process. Meas. Phenom.* **2010**, *28* (3),  
603 C4D11–C4D17. doi:10.1116/1.3368597.
- 604 42. Schwartz, G.; Riedel, C.; Arinero, R.; Tordjeman, P.; Alegría, A.; Colmenero, J. *Ultrami-*  
605 *croscopy* **2011**, *111* (8), 1366–1369. doi:10.1016/j.ultramic.2011.05.001.
- 606 43. Miccio, L. A.; Kummali, M. M.; Schwartz, G. A.; Alegría, Á.; Colmenero, J. *Ultramicroscopy*  
607 **2014**, *146*, 55–61. doi:10.1016/j.ultramic.2014.06.006.
- 608 44. Miccio, L. A.; Kummali, M. M.; Schwartz, G. A.; Alegría, Á.; Colmenero, J. *J. Appl. Phys.*  
609 **2014**, *115* (18), 184305. doi:10.1063/1.4875836.

- 610 45. Miccio, L. A.; Schwartz, G. A. Study of relaxation and transport processes by means of AFM  
611 based dielectric spectroscopy. In *AIP Conf. Proc.*; AIP Publishing LLC, 2014; pp 150–153.  
612 doi:10.1063/1.4876800.
- 613 46. Checa, M.; Neumayer, S. M.; Susner, M. A.; McGuire, M. A.; Maksymovych, P.; Collins, L.  
614 *Appl. Phys. Lett.* **2021**, *119* (25), 252905. doi:10.1063/5.0078034.
- 615 47. Gramse, G.; Edwards, M. A.; Fumagalli, L.; Gomila, G. *Appl. Phys. Lett.* **2012**, *101* (21),  
616 213108. doi:10.1063/1.4768164.
- 617 48. Gramse, G.; Casuso, I.; Toset, J.; Fumagalli, L.; Gomila, G. *Nanotechnology* **2009**, *20* (39),  
618 395702. doi:10.1088/0957-4484/20/39/395702.
- 619 49. Hudlet, S.; Saint Jean, M.; Guthmann, C.; Berger, J. *Eur. Phys. J. B* **1998**, *2* (1), 5–10. doi:10.  
620 1007/s100510050219.
- 621 50. Colchero, J.; Gil, A.; Baró, A. M. *Phys. Rev. B* **2001**, *64* (24), 245403. doi:10.1103/PhysRevB.  
622 64.245403.
- 623 51. Law, B. M.; Rieutord, F. *Phys. Rev. B* **2002**, *66* (3), 035402. doi:10.1103/PhysRevB.66.  
624 035402.
- 625 52. Mourran, A.; Tartsch, B.; Gallyamov, M.; Magonov, S.; Lambreva, D.; Ostrovskii, B. I.;  
626 Dolbnya, I. P.; de Jeu, W. H.; Moeller, M. *Langmuir* **2005**, *21* (6), 2308–2316. doi:10.1021/  
627 la048069y.
- 628 53. Magonov, S.; Alexander, J. *Beilstein J. Nanotechnol.* **2011**, *2* (1), 15–27. doi:10.3762/bjnano.  
629 2.2.
- 630 54. Sader, J. E.; Chon, J. W. M.; Mulvaney, P. *Rev. Sci. Instrum.* **1999**, *70* (10), 3967–3969. doi:  
631 10.1063/1.1150021.
- 632 55. Labuda, A.; Kocun, M.; Lysy, M.; Walsh, T.; Meinhold, J.; Proksch, T.; Meinhold, W.; Ander-  
633 son, C.; Proksch, R. *Rev. Sci. Instrum.* **2016**, *87* (7), 073705. doi:10.1063/1.4955122.

634 56. Reichel, P. Tip-sample capacitance in electrostatic force microscopy. Bachelor thesis, Univer-  
635 sity of Mainz, 2021.